



## Letter

TiO<sub>2</sub>–g-C<sub>3</sub>N<sub>4</sub> composite materials for photocatalytic H<sub>2</sub> evolution under visible light irradiation

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## ARTICLE INFO

## Article history:

Received 22 July 2010

Received in revised form

25 September 2010

Accepted 28 September 2010

Available online 8 October 2010

## Keywords:

Semiconductors

Photocatalysis

Titanium dioxide

g-C<sub>3</sub>N<sub>4</sub>

## ABSTRACT

In this investigation, we report the preparation of TiO<sub>2</sub>–g-C<sub>3</sub>N<sub>4</sub> composite materials with varying the wt.% of g-C<sub>3</sub>N<sub>4</sub>, the characterization of these materials by various techniques and photocatalytic hydrogen production under visible light irradiation in the presence of methanol. The X-ray powder diffraction (XRD) shows that the composite materials consist of anatase TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>. Fourier transform infrared (FT-IR) spectra show that the absorbance band intensity of composite materials was stronger than that of C<sub>3</sub>N<sub>4</sub>. The UV–vis absorption spectra show that the absorption edge of the composite materials shifts to the lower energy region comparing to pure anatase and to longer wavelengths with increasing the amount of C<sub>3</sub>N<sub>4</sub>. The significant photoluminescence quenching was observed in TiO<sub>2</sub>–C<sub>3</sub>N<sub>4</sub> composite materials, indicating the charge transfer from C<sub>3</sub>N<sub>4</sub> to TiO<sub>2</sub>. The visible light induced H<sub>2</sub> evolution rate was remarkably enhanced by coupling TiO<sub>2</sub> with C<sub>3</sub>N<sub>4</sub>.

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## 1. Introduction

Photocatalytic water splitting to produce hydrogen has become an area of great research interest because of the global energy crisis and environmental pollution since Fujishima and Honda first reported the decomposition of water on illuminated TiO<sub>2</sub> electrodes in 1972 [1]. To date, a variety of photocatalysts constructed from transition-metal ions with d<sub>0</sub> electronic configuration or post-transition-metal ions of d<sub>10</sub> configuration, along with group VA or VIA ions as counter-anion components [2–6] have been developed. Among various kinds of photocatalysts, TiO<sub>2</sub> is widely used due to its optical and electronic properties, long-term stability, low cost, and nontoxicity. However, TiO<sub>2</sub> could only respond to UV irradiation that takes up ~4% of solar energy, which limits the application to a great extent. In the past decades, numerous methods have been developed to increase the photocatalytic efficiency of TiO<sub>2</sub>, such as doping of TiO<sub>2</sub> with metallic (Cr, Fe, V) [7,8], and nonmetallic elements (N, C, B) [9,10] to increase the visible light absorbance, and coupling with other semiconductors [11–13] to increase the separation efficiency of photogenerated electron–hole pairs during photocatalysis.

Recently, Wang et al. [14] reported that a polymer semiconductor on the basis of a defective graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) possesses the performance of hydrogen or oxygen production from water splitting under visible light irradiation in the presence of

sacrificial donor or acceptor. The optical band gap of this polymer semiconductor was determined to be 2.7 eV. The g-C<sub>3</sub>N<sub>4</sub> photocatalyst is considered to be stable under light irradiation in water solution as well as in acid (HCl, pH 0) or base (NaOH, pH 14) solutions due to the strong covalent bonds between carbon and nitride atoms. Very recently, a composite photocatalyst of g-C<sub>3</sub>N<sub>4</sub> and TaON with visible light response was prepared by a milling-heat treatment method and was used for photodegradation of rhodamine B [15]. However, the photocatalytic activity is still low. Therefore it is expected to further improve the photocatalytic performance of g-C<sub>3</sub>N<sub>4</sub> for obtaining more effective catalyst.

In this study, the preparation of TiO<sub>2</sub>–g-C<sub>3</sub>N<sub>4</sub> composite materials with varying the wt.% of g-C<sub>3</sub>N<sub>4</sub>, characterization of these materials by various techniques and photocatalytic hydrogen production under visible light irradiation in the presence of methanol are reported.

## 2. Experimental

All starting materials were purchased from commercial sources and were used without further purification.

The TiO<sub>2</sub> powders were prepared as follows: an aqueous solution of TiCl<sub>4</sub> (200 mL, 0.5 mol dm<sup>−3</sup>) was added slowly to ammonia solution (200 mL, 2 mol dm<sup>−3</sup>) under vigorous stirring. After stirred for 30 min the resulting slurry was filtered, washed with water and ethanol subsequently, filtered again, and kept under vacuum at 400 K for 12 h. Then the powders were calcined at 650 °C in air for 2 h.

The g-C<sub>3</sub>N<sub>4</sub> was prepared by directly heating melamine under a flow of Ar gas. A quartz tube, 30 mm in inner diameter and 1000 mm in length, was used as a reaction chamber. In the synthesis, analytical grade melamine powder (10 g) was pressed into a semiclosed cylindrical quartz tube and then placed in the middle region of the reaction chamber. The quartz tube was heated to 380 °C in 5 min and then heated to 600 °C in a muffle furnace for 4 h at a heating rate of 1 °C min<sup>−1</sup> in the flow of Ar

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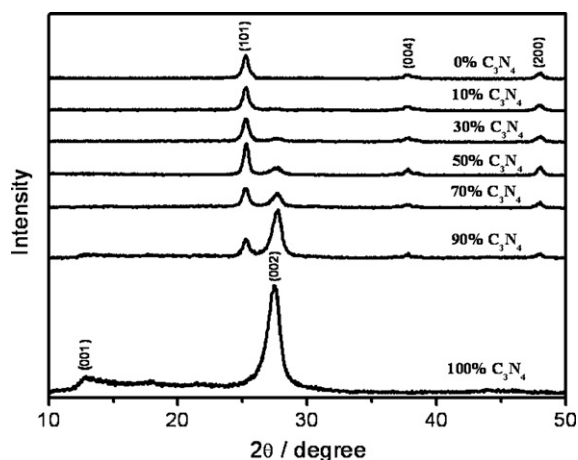


Fig. 1. The powder XRD patterns of  $C_3N_4$ ,  $TiO_2$  and  $TiO_2-C_3N_4$  composite.

gas. After the reaction, the quartz tube was cooled down to room temperature in the flow of Ar gas.

The  $TiO_2$  and  $C_3N_4$  powder were mixed by ball-milling and then calcined at  $400^\circ C$  for 1 h in a muffle furnace.

The structural properties of the materials were analyzed by X-ray powder diffraction (XRD) on a X-Pert Pro diffractometer with  $CuK\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) at a scanning speed of  $4^\circ \text{ min}^{-1}$ . FT-IR measurements were performed using a Fourier transform infrared (FT-IR) spectrometer (Nicolet 670) only in the mid infrared range of the instrument ( $400\text{--}4000 \text{ cm}^{-1}$ ) for samples dispersed in KBr pellets in 1:99 ratio. UV–vis diffuse reflection spectra were measured using a UV–vis spectrophotometer (UV2100, Shimadzu) and converted from reflection to absorbance by the Kubelka–Munk method. The photoluminescence measurements were performed in a luminescence spectrophotometer (Hitachi F-7000) operated at room temperature. The morphology of the composite materials was characterized using JEM-100CX Transmission Electron Microscopy (TEM, JEOL, Japan).

The photocatalytic reaction under visible light irradiation was carried out in an inner irradiation quartz cell (500 mL). The reaction cell was connected to a closed gas circulation system and the gases evolved were analyzed with an on-line TCD gas chromatograph (SPSIC, GC-102AT, argon carrier). In a typically photocatalytic reaction, 0.3 g of  $C_3N_4-TiO_2$  powder was suspended in 400 mL of aqueous methanol solution ( $H_2O: CH_3OH = 7:1$ , in volume). 0.5 wt.% Pt cocatalyst was loaded on the photocatalyst powder by an in situ photodeposition method to promote  $H_2$  production: under light irradiation, an equivalent molar amount of  $H_2PtCl_6$  in solution was reduced to the metallic state and deposited onto the surface of the catalyst, forming the Pt-loaded catalyst. A 450 W high-pressure mercury lamp was blocked by a  $NaNO_2$  aqueous solution (2 M) filled in quartz tube. In this case, it is considered that an emission of a high-pressure mercury lamp at 436 nm mainly contributes for the reaction.

### 3. Results and discussion

Fig. 1 shows the powder XRD patterns of  $C_3N_4$ ,  $TiO_2$  and  $TiO_2-C_3N_4$  composite. Two pronounced peaks were found in  $g-C_3N_4$  at  $27.4^\circ$  and  $13.1^\circ$ , which can be indexed to (002) and (001) diffraction planes of the graphite-like carbon nitride [16] and cor-

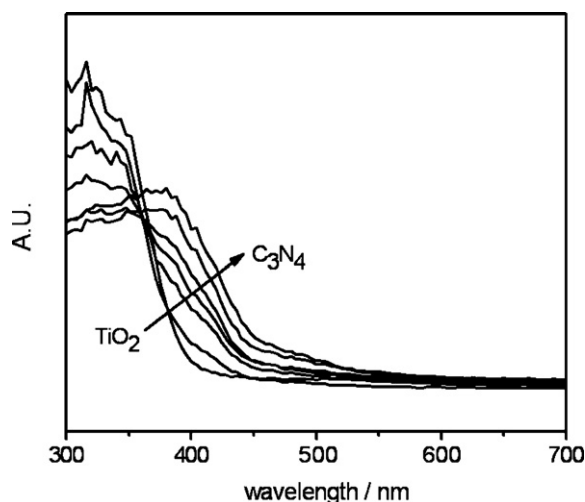


Fig. 3. The UV–vis absorption spectra of  $C_3N_4$ ,  $TiO_2$  and  $TiO_2-C_3N_4$  composite materials.

respond to the characteristic interplanar staking peak of aromatic systems and the inter-layer structural packing, respectively. No impurity phase was observed in the  $TiO_2$  sample, which is consistent with the XRD pattern of anatase. The  $TiO_2-C_3N_4$  sample presents a two-phase composition:  $C_3N_4$  and  $TiO_2$ .

Fig. 2a and b shows the Fourier transform infrared (FT-IR) spectra of  $C_3N_4$  and  $TiO_2-50 \text{ wt.}\% C_3N_4$  respectively. The FT-IR spectra of the synthesized  $C_3N_4$  show the features very similar to those of published results [17]. The absorption band near  $1640 \text{ cm}^{-1}$  is attributed to C–N stretching, while the three at  $1240$ ,  $1320$  and  $1403 \text{ cm}^{-1}$  to aromatic C–N stretching. The band near  $808 \text{ cm}^{-1}$  is attributed to out-of plane bending modes of C–N heterocycles. A broad band near  $3140 \text{ cm}^{-1}$  corresponds to the stretching modes of terminal  $NH_2$  or  $NH$  groups at the defect sites of the aromatic ring [18]. The FT-IR spectra of  $TiO_2-50 \text{ wt.}\% C_3N_4$  is similar to those of  $C_3N_4$ . However, the absorbance band intensity of  $TiO_2-50 \text{ wt.}\% C_3N_4$  is stronger than that of  $C_3N_4$ , which indicates the formation of a composite between  $TiO_2$  and  $C_3N_4$ .

The optical properties of the samples were investigated by UV–vis diffuse reflectance spectroscopy. Fig. 3 shows the UV–vis absorption spectra of  $C_3N_4$ ,  $TiO_2$  and  $TiO_2-C_3N_4$  composite materials. The absorption edge of the  $TiO_2$  sample occurs at ca. 390 nm, and the band gap energy is estimated to be about 3.18 eV. After coupled with  $C_3N_4$ , the absorption edge shifts to the lower energy region. We can also see that the absorption edges of the composite samples shift remarkably to longer wavelengths with increasing the amount of  $C_3N_4$ . The decrease in band gaps of the samples is

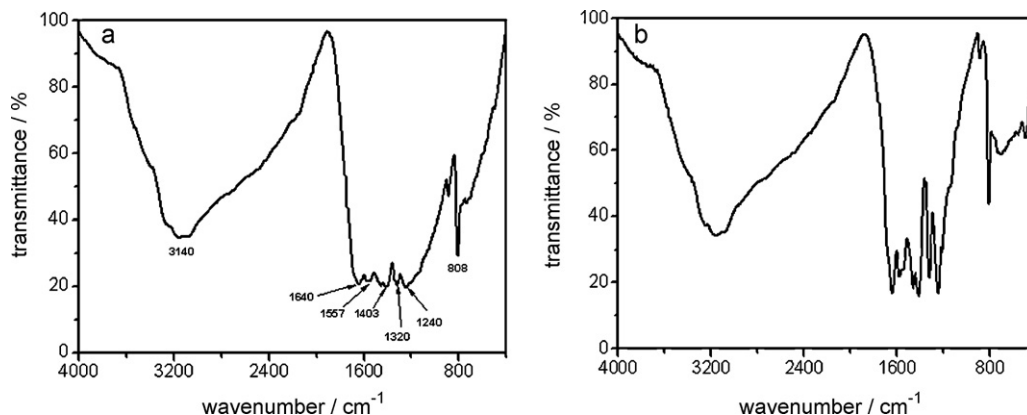


Fig. 2. The Fourier transform infrared (FT-IR) spectra of  $C_3N_4$  (a) and  $TiO_2-50 \text{ wt.}\% C_3N_4$  (b).

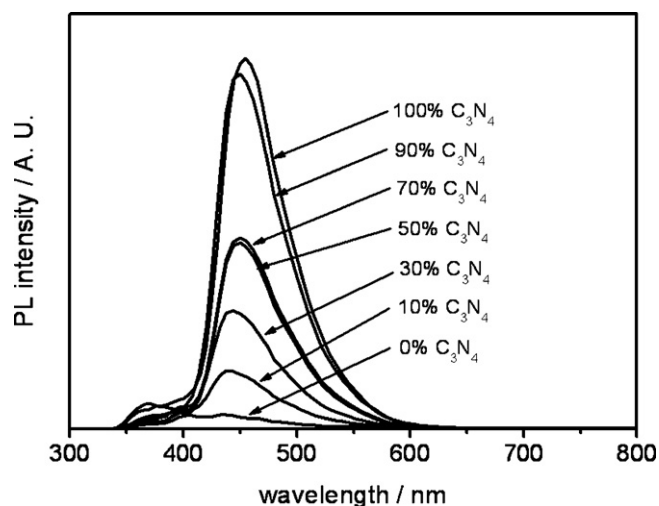


Fig. 4. The photoluminescence spectra of  $C_3N_4$ ,  $TiO_2$  and  $TiO_2-C_3N_4$  composite materials.

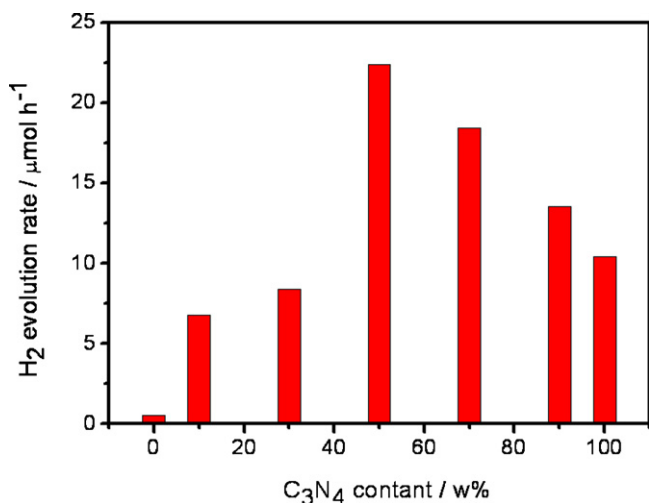


Fig. 5. The average hydrogen production rates as a function of wt.%  $C_3N_4$  from the first 3 h of the reaction duration.

from 3.18 of  $TiO_2$  to 2.68 eV of  $C_3N_4$  when the content of  $C_3N_4$  is increased from 0% to 100%.

Fig. 4 shows the photoluminescence (PL) spectra of the  $TiO_2-C_3N_4$  composite materials. The peak at 453 and 369 nm could be related to electron-hole recombination of  $C_3N_4$  and  $TiO_2$ , respectively. The significant PL quenching was observed in  $TiO_2-C_3N_4$

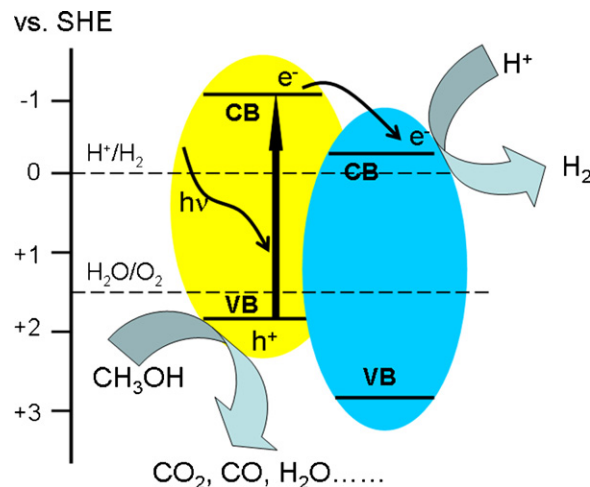


Fig. 7. The scheme for visible light-driven electron-hole separation of the  $TiO_2-C_3N_4$  composite materials.

composite materials as the content of  $TiO_2$  increases. The quenching is due to the charge transfer occurs from  $C_3N_4$  to  $TiO_2$ .

Photocatalytic hydrogen evolution from aqueous solutions containing  $CH_3OH$  ( $H_2O:CH_3OH = 7:1$ , in volume) was conducted under visible light using a 450 W high-pressure mercury lamp blocked by a  $NaNO_2$  aqueous solution (2 M). Fig. 5 shows the average hydrogen production rates as a function of wt.%  $C_3N_4$  from the first 3 h of the reaction duration. No hydrogen produced under the dark condition and without catalyst in presence of light. The evolution of hydrogen commenced after switching the lamp on in presence of catalyst only. However, the visible light induced  $H_2$  evolution over  $TiO_2$  is negligible. As shown in Fig. 5, the visible light induced  $H_2$  evolution rate was remarkably enhanced by coupling  $TiO_2$  with  $C_3N_4$ . Sample of  $TiO_2-50$  wt.%  $C_3N_4$  exhibits the highest activity with a hydrogen production rate of  $22.4 \mu\text{mol h}^{-1}$ . The visible light induced  $H_2$  evolution rate of  $TiO_2-50$  wt.%  $C_3N_4$  was 2 times of those of pure  $C_3N_4$ .

The morphology of  $TiO_2-50$  wt.%  $C_3N_4$  composite materials before and after photocatalytic reaction was investigated using TEM, as shown in Fig. 6(A) and (B) respectively. The particle size of the composite materials was estimated to be in the range of approximately 30–40 nm. After photocatalytic reaction, Pt nanoparticles with an average diameter of approximately 3–5 nm were deposited on the surface of the composite, which act as  $H_2$  evolution site during photocatalytic reaction.

The enhancement of photocatalytic performance of the composite materials is attributed mainly to the effectively separation of photogenerated electron-hole pairs. Fig. 7 shows the scheme for visible light-driven electron-hole separation of the  $TiO_2-C_3N_4$

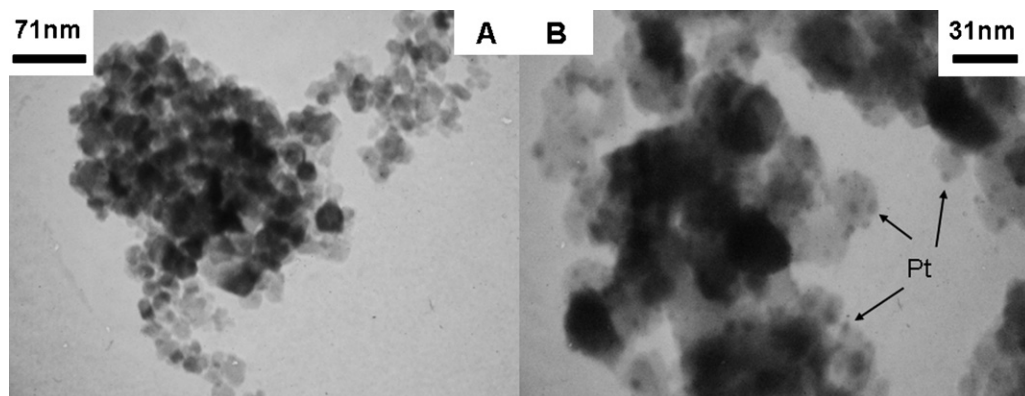


Fig. 6. The TEM images of  $TiO_2-50$  wt.%  $C_3N_4$  composite materials before (A) and after (B) photoreaction.

composite materials. The conduction band (CB) and valence band (VB) potentials of the  $\text{TiO}_2$  and  $\text{g-C}_3\text{N}_4$  are also shown in Fig. 7. The CB and VB edge potentials of  $\text{g-C}_3\text{N}_4$  are at  $-1.12$  and  $1.57$  eV, respectively [14]. The CB and VB edge potentials of  $\text{TiO}_2$  are at  $-0.29$  and  $2.91$  eV, respectively [19]. Since the CB edge potential of  $\text{g-C}_3\text{N}_4$  ( $-1.12$  eV) is more negative than that of  $\text{TiO}_2$  ( $-0.29$  eV) the photoinduced electrons on  $\text{g-C}_3\text{N}_4$  particle surfaces transfer easily to  $\text{TiO}_2$  via the well developed interface and reduce  $\text{H}^+$  to  $\text{H}_2$ . However, the photoinduced holes were suspended in  $\text{g-C}_3\text{N}_4$  due to the large difference in VB edge potentials and oxidize the sacrificial agent ( $\text{CH}_3\text{OH}$ ) to products, such as  $\text{CO}_2$ ,  $\text{CO}$ .

#### 4. Conclusions

In summary, the  $\text{TiO}_2$ - $\text{g-C}_3\text{N}_4$  composite materials with varying the wt.% of  $\text{g-C}_3\text{N}_4$  were prepared in this investigation. The composite materials present a two-phase composition: anatase  $\text{TiO}_2$  and  $\text{g-C}_3\text{N}_4$ . The absorbance band intensity of composite materials is stronger than that of  $\text{C}_3\text{N}_4$ . The absorption edge of the coupling materials shifts to the lower energy region comparing to pure anatase and to longer wavelengths with increasing the amount of  $\text{C}_3\text{N}_4$ . The photocatalytic result show that the visible light induced  $\text{H}_2$  evolution rate was remarkably enhanced by coupling  $\text{TiO}_2$  with  $\text{C}_3\text{N}_4$  due to the effectively separation of photogenerated electron-hole pairs.

#### Acknowledgments

We thank the Fundamental Research Funds for the Central Universities (Project No. 2009SCU11105) and Prof. Can Li for the financial support.

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